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PROPERTIES OF NEW SUPERCONDUCTING MATERIALS.(U)  
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FINAL REPORT

FOR

AIR FORCE OFFICE OF SCIENTIFIC RESEARCH

AIR FORCE SYSTEMS COMMAND, U.S. AIR FORCE

Grant No. AFOSR-73-2435

PROPERTIES OF NEW SUPERCONDUCTING MATERIALS

by

Professor T. H. Geballe  
Principal Investigator

Department of Applied Physics  
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Stanford, California 94305

G. L. Report No. 2760

December 1977

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U.S. Government Printing Office

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20. For example, a new method of stabilizing films of  $Nb_3Ge$  with superconducting transitions above 22K enabled the homogeneous stoichiometric composition to be reached by epitaxial growth on polycrystalline substrates. Thus, new approaches have been uncovered for synthesizing compounds with even higher superconducting transition temperatures.

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During the period of the grant and its predecessor an entirely new class of superconductors - the intercalated layered transition metal dichalcogenides - was discovered and investigated. More than 75 new superconducting crystals of this class have been synthesized in collaboration with the industry-supported group of F. R. Gamble and associates. This probably represents most of the new superconductors discovered during the past 10 years.

The distinguishing feature of these intercalated superconductors is that there are easily identifiable dichalcogenide layers in which the conduction takes place. These superconductors are the most anisotropic compounds known. The conductivity in the approximately 6 angstrom thick dichalcogenide layer was found to be that of a typical metallic alloy. The rest of the unit cell of the crystal is formed by an organic insulating layer which, in the case of octadecylamine is 50 angstroms thick. Electrical anisotropies of greater than  $10^5$  have been found at room temperature for some of the more perfect crystals. The anisotropic Ginzburg-Landau parameters in 20 different single crystal superconductors were investigated using newly developed heat capacity methods as well as electrically. The coherence length in  $\text{TaS}_{1.6}\text{Se}_{1.4}(\text{C}_6\text{H}_5\text{N})$  was found to be only 4.5 angstroms in the direction perpendicular to the layers (vs 220 angstroms parallel to the layers). The perpendicular distance is less than the interlayer separation of the conducting dichalcogenide layers demonstrating that the magnetic field can pass through the superconductor without inducing large diamagnetic screening currents. As a consequence, the upper critical field (measured parallel to the surface) was found to increase at the record rate of  $> 140,000$  gauss per degree as the sample was cooled below its transition.

Using new electrolytic methods of intercalations, the stoichiometric compound  $-\text{TaS}_2(\text{H}_2\text{N}-\text{CH}_2-\text{CH}_2-\text{NH}_2)_{1/4}$  was formed upon intercalation of

monocrystalline  $4H(b)\text{-TaS}_2$  (a polymorph of  $\text{TaS}_2$  with four layers per unit cell), with ethylenediamine. In contrast to  $4H(b)\text{-TaS}_2$  ( $T_c < 0.5\text{K}$ ), the intercalated compounds were found by heat capacity measurements (1) to undergo a bulk superconducting transition near 3 K, (2) to have a marked increase in its electronic density of states and (3) to show the intriguing possibility of being a superconductor with no energy gap. The transition temperature was also raised to about 3 K in  $4Hb\text{-TaS}_2$  by intercalation of hydrogen apparently because of hydrogen's role in suppressing the charge density waves. The latter are instabilities driven by the Fermi-surface. Thermal and electrical properties in the hexagonal polymorphs of  $\text{NbSe}_2$ ,  $\text{TaS}_2$ ,  $\text{TaSe}_2$ , and the mixed trigonal and hexagonal polymorphs of  $\text{TaS}_2$  and  $\text{TaSe}_2$  were measured and interpreted in terms of a model in which only a small fraction of the Fermi surface participates in the formation of the charge density wave. A systematic study of the optical properties in a series of crystals in the  $1T(\text{Ti}_{1-x}\text{Ta}_x\text{S}_2)$  system has suggested why these compounds behave electrically as semiconductors on the Ta-rich side rather than as the superconductors one would expect. Analysis of their Drude-like optical behavior shows that scattering times become remarkably short for the tantalum rich compounds ( $x \geq 0.6$ ), so short in fact that the Boltzmann transport equation no longer applies. An unexpected quadratic dependence of resistivity upon temperature was found over a wide range in  $\text{TiS}_2$  suggesting electron-hole scattering as the mechanism.

B. W. Roberts in "Survey of Superconductive Materials and Critical Evaluation of Selected Properties", has tabulated in Table III, a copy of which is appended, the properties of superconducting materials with organic and related constituents. References 1128 and 1192 in particular refer to research discussed above. This important work has resulted in widespread national and international recognition and has directly stimulated new



research in other laboratories. A special conference was sponsored by the ONR, "Physics and Chemistry of Layered Compounds", Monterey, California, August 1972, a Gordon Research Conference has been held "Chemistry and Physics of Solids", Holderness School, Plymouth, N.H., August 1975, and an international conference, "Layered Semiconductors and Metals", a Satellite Conference of I.C.P.S., has been held in Bari, Italy, September 1976. An active group has developed at the University of Illinois under F. C. Brown, W. M. McMillan, and others. Work in the high magnetic field properties and superconductive fluctuations at high temperatures was undertaken by M. R. Beasley, A. H. Luther and R. A. Klemm at Harvard and has been continued by D. Prober at Yale. Important charge density wave instabilities were discovered at Bell Laboratories by F. J. DiSalvo, J. R. Wilson and S. Mahajan. More indirectly linked research on the mechanism of the intercalation reaction has resulted in promising new battery electrodes by workers at Exxon and Bell.

During the latter part of the grant an improved method of growing high temperature A-15 compounds was developed using the method of electron beam codeposition. Elements were evaporated from individually controlled electron beam sources exploiting a method of fast feedback developed in this laboratory. The phase diagram of  $\text{Nb}_3\text{Ge}$  below 1000 K was investigated and showed that the Ge-rich boundary extends beyond the 20-22 at % previously established and exhibits an unusual temperature dependence when trace gases such as  $\text{O}_2$  and  $\text{Cl}_2$  are present in the evaporator at partial pressures of  $10^{-5}$  -  $10^{-6}$  torr. A careful investigation of  $\text{Nb}_3\text{Ge}$  showed no evidence for a low-temperature martensitic transformation such as that found in  $\text{Nb}_3\text{Sn}$ . A potentially very powerful new method, "polycrystalline" epitaxy, which in some sense is equivalent to molecular beam epitaxy, was developed for synthesizing metastable compounds. Using this method the Ge-rich phase



boundary of  $\text{Nb}_3\text{Ge}$  was extended from the thermodynamic limit of 19 atomic percent to 26 at %. Either  $\text{Nb}_3\text{Ir}$  or  $\text{Nb}_3\text{Rh}$  pre-evaporated as a polycrystalline substrate was found to be effective since both have suitable lattice constants. The resulting  $\text{Nb}_3\text{Ge}$  films were found to have superconducting transition temperatures which maximize, at the stoichiometric composition, with temperatures in excess of 22 K. Another consequence of this electron beam technology was the development of a method to produce oxide tunnel junctions of  $\text{Nb}_3\text{Sn}$  in which parameters such as composition could be varied systematically for the first time.

New approaches have been opened for the synthesis of stoichiometric compositions of other A-15 compounds for which the thermodynamic limit of the B element is less than 25% such as  $\text{Nb}_3\text{Al}$  and  $\text{Nb}_3\text{Ga}$ . By analogy with what has been found for  $\text{Nb}_3\text{Ge}$  the superconducting transitions should be above the present values of 18.8 K and 20.7 K respectively. The possibility of making  $\text{Nb}_3\text{Si}$  stable in the A-15 phase by epitaxial means has also been opened, and here the transition temperature is expected to exceed the highest present day value of 23 K.

There were other significant developments in this 5 year period including a new method of introducing hydrogen into palladium electrolytically from methanol at - 80 C which resulted in stoichiometric  $\text{PdH}$  with  $T_c = 9.1$  K. This method has been adopted by other research groups both in the US and abroad. A relationship was established between hydrogen concentration and the resistivity maxima observed at temperatures above  $T_c$  due to clustering of the hydrogen. The prototype of a possible new class of metallic compounds was found in collaboration with C. W. Chu, visiting from Cleveland State, and A. P. Rusakov, a Russian visitor at Stanford under an exchange program.

CuCl was found to transform into a highly-conducting state over a relatively narrow pressure range above 40 Kbar. The new phase occurred as an intermediate between the low pressure (cubic) and high pressure (rock-salt) phases both of which are insulators. Further research has been carried out by both Dr. C. W. Chu, now at University of Houston, and Dr. Rusakov in Moscow.

TABLE 3. Properties of Superconductive Materials with Organic and Related Constituents

NOTE: "HF" Signifies high-magnetic-field data in Table 5.

Material	$T_c$ (K)	$H_0$ (oersted)	Crystal Structure	$T_n$ (K)	Refs.
Al(and tetracyanoquino- dimethan)	2.7-5.24 1.9-3.7 (annealed)				V1078
Be(with KCl layers; deposit 4.2K)	10.6-6.5				V1028
Be(with zinc-etioporphyrin; deposit 4.2K; $\geq 500 \text{ \AA}$ )	10.2				V1028
$\text{CaH}_{18}\text{N}_4\text{C}$				1.9	010
$\text{H}_{12}\text{LiN}_4$				1.9	010
In (with Anthraquinone, 5000 $\text{\AA}$ )	3.4-4.6				V1076 V1528
$\text{MoS}_2\text{Ba}_{0.2}(\text{NH}_3)_x$	5.7		HEX		1918
$\text{MoS}_2\text{Ca}_{0.2}(\text{NH}_3)_x$	3.6		HEX		1918
$\text{MoS}_2\text{Sr}_{0.01-1}(\text{NH}_3)_{0.01-1.62}$	5.2-4.9				1918
$\text{MoS}_2\text{Yb}_{0.1}(\text{NH}_3)_{0.16}$	2.4		HEX		1918
$\text{MoSe}_2\text{Sr}_{0.2}(\text{NH}_3)_x$	5.0		HEX		1918
NS	0.26				1986 1975#
$\text{NbS}_2$ (Ammonia)	2.0		HEX		1192
$\text{NbS}_2$ (Aniline) $_7$	4.0		HEX		1192
$\text{NbS}_2$ (S-collidine) $_{0.17}$	3.5		HEX		1192
$\text{NbS}_2$ (pyridine) $_{0.5}$	4.0		HEX		1192 1027
$\text{NbS}_2$ (tributylphosphine) $_{0.125}$	3.5		HEX		1192
$\text{Nb}_3\text{Sn}$ (with $\text{CO}_2$ , $\text{CO}$ , $\text{CH}_3$ , $\text{N}_2$ , $\text{O}_2$ , ammonium, boron trichloride, ethane, hydrogen sulfide, nitrogen oxide, propane)		HF			1169 1168 V1437
$\text{PdTe}_2$ (pyridine) $_{0.5}$	1.65				1027
$\text{S}_{2-1}\text{Se}_{0-1}\text{Ta}$ (pyridine) $_{0.5}$	0.8-3.3-1.6		HEX		1910
$\text{S SeTa}$ (pyridine)	1.5	HF	HEX		1262
$\text{S}_2\text{Ta}$ (2-aminopyridine) $_{0.53}$	3.25		HEX		1128
$\text{S}_2\text{Ta}$ (4-aminopyridine) $_{0.51}$	3.4		HEX		1128
$\text{S}_2\text{Ta}$ (ammonia)	4.2		HEX		1192
$\text{S}_2\text{Ta}$ (ammonium acetate)	2.0		HEX		1192
$\text{S}_2\text{Ta}$ (ammonium hydroxide)	3.3		HEX		1192
$\text{S}_2\text{Ta}$ (amylamine)	2.2				1192
$\text{S}_2\text{Ta}$ (aniline)	3.1		HEX		1192
$\text{S}_2\text{Ta}$ (aniline) $_{0.75}$	3.1		HEX		1192



TABLE 3 (Cont'd). Properties of Superconductive Materials with Organic and Related Constituents

NOTE: "HF" Signifies high-magnetic-field data in Table 5.

Material	$T_c$ (K)	$H_0$ (oersted)	Crystal Structure	$T_n$ (K)	Refs.
$S_2Ta$ (barium hydrate) <sub>0.15</sub>	3.74	150	HEX		1845
$S_2Ta$ (butylamine)	2.5		HEX		1192
$S_2Ta$ (butyramide)	3.1		HEX		1192
$S_2Ta$ (calcium <sub>(0.3)</sub> hydrate)	3.47				1770 1845
$S_2Ta$ (calcium <sub>(0.15)</sub> hydrate)	3.47	130	HEX		1845
$S_2Ta$ (cesium <sub>(0.3)</sub> hydrate)	2.75, 2.80	110	HEX		1845 1770
$S_2Ta$ (cesium hydroxide)	3.8		HEX		1192
$S_2Ta$ (s-collidine) <sub>0.17</sub>	2.0, 1.95		HEX		1192 1871
$S_2Ta$ (2,6-diaminopyridine) <sub>0.53</sub>	3.50		HEX		1128
$S_2Ta$ (2-dimethylamino-pyridine) <sub>0.32</sub>	3.15		HEX		1128
$S_2Ta$ (4-dimethylamino-pyridine) <sub>0.34</sub>	2.30		HEX		1128
$S_2Ta$ (N,N-dimethylaniline)	4.3		HEX		1192
$S_2Ta$ (2,6-dimethylpyridine) <sub>0.20</sub>	2.15		HEX		1128
$S_2Ta$ (4,4'-dipyridyl)	2.5		HEX		1192
$S_2Ta$ (ethylamine)	3.3		HEX		1192
$S_2Ta$ (2-ethylpyridine) <sub>0.29</sub>	3.0		HEX		1128
$S_2Ta$ (3-ethylpyridine) <sub>0.29</sub>	4.50		HEX		1128
$S_2Ta$ (4-ethylpyridine) <sub>0.33</sub>	2.95		HEX		1128
$S_2Ta$ (hexanamide)	3.1		HEX		1192
$S_2Ta$ (hydrazine)	4.7		HEX		1192
$S_2Ta$ (hydrogen) <sub>0-0.87</sub>	0.8-4.2-<0.5				1871
$S_2Ta$ (2-isopropylpyridine) <sub>0.25</sub>	3.80		HEX		1128
$S_2Ta$ (4-isopropylpyridine) <sub>0.28</sub>	2.82		HEX		1128
$S_2Ta$ (isoquinoline)	2.5		HEX		1192
$S_2Ta$ (lithium hydrate) <sub>0.3</sub>	3.83	170	HEX		1845
$S_2Ta$ (lithium hydroxide)	4.5		HEX		1192
$S_2Ta$ (methylamine)	4.2		HEX		1192
$S_2Ta$ (2-methylpyridine) <sub>0.34</sub>	2.95		HEX		1128
$S_2Ta$ (3-methylpyridine) <sub>0.33</sub>	2.95		HEX		1128
$S_2Ta$ (4-methylpyridine) <sub>0.33</sub>	2.70		HEX		1128
$S_2Ta$ (octadecylamine)	3.0		HEX		1192

TABLE 3 (Cont'd). Properties of Superconductive Materials with Organic and Related Constituents

NOTE: "HF" Signifies high-magnetic-field data in Table 5.

Material	$T_c$ (K)	$H_0$ (oersted)	Crystal Structure	$T_n$ (K)	Refs.
$S_2Ta$ (pentadecylamine)	2.8		HEX		1192
$S_2Ta$ (p-phenylenediamine)	3.3		HEX		1192
$S_2Ta$ (p-phenylenediamine) <sub>0.25</sub>	2.9		HEX		1192
$S_2Ta$ (2-phenylpyridine) <sub>0.255</sub>	3.15		HEX		1128
$S_2Ta$ (4-phenylpyridine) <sub>0.26</sub>	1.6		HEX		1128
$S_2Ta$ (picoline) <sub>0.34</sub>	2.70		HEX		1871
$S_2Ta$ (potassium formate)	4.7		HEX		1192
$S_2Ta$ (potassium (0.3) hydrate)	5.25	230	HEX		1845 1770
$S_2Ta$ (potassium hydroxide)	5.3		HEX		1192
$S_2Ta$ (propylamine)	3.0		HEX		1192
$S_2Ta$ (4-propylpyridine) <sub>0.25</sub>	2.75		HEX		1128
$S_2Ta$ (2-propylpyridine) <sub>0.245</sub>	2.85		HEX		1128
$S_2Ta$ (pyridine) <sub>0.5</sub>	3.5	HF	HEX		1192 1027
$S_2Ta$ (pyridine) <sub>0.5</sub>	3.55		HEX		1128 1871
$S_2Ta$ (pyridine) <sub>0.5</sub>	3.25	HF			1262 1430
$S_2Ta$ (pyridine-N-oxide)	2.5		HEX		1192
$S_2Ta$ (pyridinium chloride)	3.1		HEX		1192
$S_2Ta$ (quinoline)	2.8		HEX		1192
$S_2Ta$ (rubidium (0.3) hydrate)	4.40	210	HEX		1845 1770
$S_2Ta$ (rubidium hydroxide)	4.3		HEX		1192
$S_2Ta$ (septadecylamine)	2.7		HEX		1192
$S_2Ta$ (sodium (0.3) hydrate)	5.41	250	HEX		1845 1770
$S_2Ta$ (sodium hydroxide)	4.8		HEX		1192
$S_2Ta$ (stearamide)	3.1, 3.0		HEX		1192
$S_2Ta$ (strontium (0.2) ammonium)	2.8		HEX		1918
$S_2Ta$ (strontium (0.15) hydrate)	4.03	190	HEX		1845
$S_2Ta$ (tetradecylamine)	2.4		HEX		1192
$S_2Ta$ (N,N,N',N'-tetramethyl-p-phenylene-diamine)	2.9		HEX		1192
$S_2Ta$ (thiobenzamide)	3.3		HEX		1192
$S_2Ta$ (tributylamine)	3.0		HEX		1192
$S_2Ta$ (tributylphosphine) <sub>0.125</sub>	2.0		HEX		1192
$S_2Ta$ (tridecylamine)	2.5		HEX		1192

TABLE 3 (Cont'd). Properties of Superconductive Materials with Organic and Related Constituents

NOTE: "Hf" Signifies high-magnetic-field data in Table 5.

Material	$T_c$ (K)	$H_0$ (oersted)	Crystal Structure	$T_n$ (K)	Refs.
$S_2Ta$ (2,4,6-trimethyl- pyridine) <sub>0.165</sub>	1.95		HEX		1128
$S_2Ta$ (triton B)	5.0		HEX		1192
$S_2Ta$ (valeramide)	2.9		HEX		1192
$S_2Ta_{0.8}W_{0.2}$ (s-collidine) <sub>0.17</sub>	2.0		HEX		1192
$S_2Ta_{0.3}W_{0.7}$ (s-collidine) <sub>0.17</sub>				~0.4	1192
$S_2Ti$ (ammonia)			HEX	0.3	1192
$S_2Ti$ (aniline)			HEX	0.3	1192
$S_2Ti$ (s-collidine) <sub>0.17</sub>			HEX	0.3	1192
$S_2Ti$ (pyridine) <sub>0.5</sub>			HEX	0.3	1192
$S_2Ti$ (tributylphosphine) <sub>0.125</sub>			HEX	0.3	1192
$S_2W$ (strontium ammonium) <sub>(0.2)</sub>	3.5		HEX		1918
$S_2W$ (ytterbium ammonium) <sub>(0.4)</sub>	2.2		HEX		1918
$S_2Zr$ (ammonia)			HEX	0.3	1192
$Se_2Ta$ (pyridine) <sub>0.5</sub>	1.5		HEX		1027
$Se_2W$ (strontium ammonium) <sub>(0.2)</sub>	~1.4		HEX		1918
V (co-deposited with organic compounds, 50-200Å)	$T_c^I$ (+~0.1, -~0.1)				¶1802



# PUBLICATIONS DURING GRANT PERIOD

1. "Structure and Superconducting Properties of Intercalated  $4\text{Hb-TaS}_2$ ," by S. F. Meyer, T. H. Geballe, and J. V. Acrivos, Bull. Am. Phys. Soc. 18 384 (March 1973).
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10. "Optical and Heat Capacity Studies of the Solid Solutions  $\text{Ti Ta}_{1-x}\text{S}_2$ ," by J. A. Benda, C. N. King, K. R. Pisharody, and W. A. Phillips, Internal Memorandum (January 1974). Presented at 13th International Conference on Low Temperature Physics, LT-13, (August 1972). Published in LT-13, Vol. 4.

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17. "The Low Temperature Heat Capacity of Amorphous Germanium," by C. N. King, W. A. Phillips, and J. P. DeNeufville, Phys. Rev. Letters 32, 538 (1974).
18. "Preparation and Properties of 1T-TaSe<sub>2</sub>," by F. J. Di Salvo, B. G. Bagley, R. G. Maines, J. V. Waszczak, and R. E. Schwall, Solid State Commun. 14, 497 (1974).
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20. "Effect of Hydrogen Concentration on Superconductivity and Clustering in Palladium Hydride," by J. M. E. Harper, Phys. Letters 47A, 69 (1974).
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31. "Electron Beam Evaporation Synthesis of Al<sub>5</sub> Superconducting Compounds: Accomplishments and Prospects," by R. H. Hammond IEEE Trans. Mag. MAG-11, 201 (1975).
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37. "Detection of Nuclear Quadrupole Resonances via Induced Longitudinal Magnetization," by T. Jach, Appl. Phys. Lett. 28, 49 (1976).
38. "Superconductivity and Range of Existence of the Al<sub>5</sub> Phase in the Nb-Ge Systems," by A. Hallak, R. H. Hammond, R. B. Zubeck, and T. H. Geballe, Bull. Am. Phys. Soc. 21, 340 (1976).
39. "Low Temperature Specific Heat of Layered Compounds," by R. E. Schwall, G. R. Stewart, and T. H. Geballe, J. Low Temp. Phys. 22, 557 (1976).
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44. "Magnetic and Electric Properties of MnSb," by T. Chen, J. W. Allen, and G. R. Stewart, in Proceedings of 21st Annual Conference on Magnetism and Magnetic Material, Philadelphia, December 1975.
45. "The Elastic Behavior of 2H-TaSe<sub>2</sub>," by M. Barmatz, L. R. Testardi, F. J. Di Salvo, and J. M. E. Harper, Phys. Rev. B 13, 4637 (1976).
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48. "Properties of New Superconducting Materials," Annual Research Progress Report, (November, G. L. Report 2635).
49. "Superconducting Materials," by T. H. Geballe, presented at Distinguished Lecture Series III, University of New Mexico, March 1976 (Proceedings in press).

50. "Superconductivity Defects and Stoichiometry in Al<sub>5</sub> Materials," by R. C. Dynes, J. M. Poate, L. R. Testardi, A. R. Storm, and R. H. Hammond, IEEE Trans. Mag. MAG-13, 640 (1977).
51. "Electron-Beam Fabrication of Al<sub>5</sub> Superconducting Materials," by R. H. Hammond and R. B. Zubeck, presented at American Society for Metals Conference on Manufacture of Superconducting Materials, Port Chester, New York, November 1976, in press.
52. "Thermal Properties of Layered Transition Metal Dichalcogenides at Charge Density Wave Transitions," by J. M. E. Harper, T. H. Geballe and F. J. Di Salvo, Phys. Rev. B15, 2943 (1977).
53. "Epitaxial Growth of High-T Superconducting Nb<sub>3</sub>Ge on Nb<sub>3</sub>Ir," by A. H. Dayem, T. H. Geballe, R. B. Zubeck, A. B. Hallak and G. W. Hull, Jr., Appl. Phys. Lett. 30, 541 (1977).
54. "Epitaxial Growth of Nb<sub>3</sub>Ge on Nb<sub>3</sub>Ir and Nb<sub>3</sub>Rh," by A. H. Dayem, T. H. Geballe, R. B. Zubeck, A. B. Hallak and G. W. Hull, in press.

PERSONS WORKING ON GRANT DURING GRANT PERIOD

Benda, J. A.	Ph.D. awarded June 1973
(Thesis title: Optical and Heat Capacity Studies of Some Transition Metal Compounds)	
Bilir, N.	Ph.D. awarded May 1974
(Thesis title: Low Temperature Heat Capacities of Open-Structured Crystals)	
Early, S. R.	Ph.D. expected June 1978
Feldman, R.	Ph.D. expected June 1980
Hallak, A. B.	Ph.D. awarded June 1976
(Thesis title: Superconductivity and Range of Existence of the A15 Phase in the Niobium-Germanium System)	
Hammond, R. H.	Senior Research Associate
Harper, J. M. E.	Ph.D. awarded October 1975
(Thesis title: Thermal Properties of Metals with Low Temperature Structural Instabilities)	
Ierley, G.	Received Masters September 1977
Jach, T. J.	Ph.D. awarded August 1975
(Thesis title: Observation of Nuclear Quadrupole Resonance with Superconducting Magnetometers)	
Kimhi, D.*	Ph.D. expected June 1979
Kwo, J.*	Ph.D. expected June 1979
McDavid, G. T.	Received Masters April 1974
Meyer, S. F.	Ph.D. awarded January 1974
(Thesis title: The Effect of Intercalate and Layer Properties on Superconducting Transition Metal Dichalcogenides)	
Moore, D. F.*	Ph.D. expected April 1978
O'Connor, M. C.	Ph.D. expected June 1978
Pisharody, R.	Visiting Research Associate
Phillips, A. W.	Visiting Assistant Professor
Salem, J. R.	Research Associate



Schwall, R. E.      Ph.D. awarded June 1973

(Thesis title: Low Temperature Properties of Layered Transition  
Metal Dichalcogenide Compounds)

Stewart, G. R.      Ph.D. awarded October 1975

(Thesis title: I. Size Effects in the Heat Capacity of Small Metal  
Particles II. Heat Capacity of the Systems MnBe  
and MnSb)

Rowell, J. M.      Consulting Professor

Younge, R. C.      Graduate Research Assistant

Zubeck, R. B.      Ph.D. awarded September 1973

(Thesis title: Effects of Deformation on Mixed State Heat Capacity)

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Presentations and Visits During Grant Period

T. H. Geballe

- 10/21/74 - SLAC (seminar: "High Temperature Superconductors")
- 11/13/74 - UC Berkeley (seminar: "High Temperature Superconductors and Their Applications")
- 11/18/74 - Bell Labs (seminar: "Applied Superconductivity: Status and Prospects")
- 4/4/75 - CERL, Leatherhead, Surrey, England (visited J. Sutton and discussed use of high- $T_c$  superconductors in applied technology)
- 5/2/75 - Cambridge Univ., England (seminar: "Charge Density Waves in Superconductors and Transition Metal Dichalcogenides")
- 5/9/75 - Brown, Bovari Co., Baden, Switzerland (visited A. Menth, discussed superconducting technology)
- 5/14/75 - Institute fur Experimentelle Kernphysik, Karlsruhe, Germany (visited W. Heinz, discussed synthesis of superconducting compounds using dual beam and sputtering techniques; also, new developments of Karlsruhe in ion implantation)
- 5/15/75 - Siemens, Stuttgart, Germany (visited M. Wilhelm, discussed synthesis of  $V_3Ga$ )
- 5/29/75 - A.E.R.E., Harwell, England (visited R. Bett, J. Charlesworth, J. Lee, P. Madsen; discussed  $Nb_3Sn$  irradiation damage and manufacture of multifilamentary  $Nb_3Sn$ )
- 6/4/75 - Imperial College, England (seminar: "Instabilities in Superconductors")
- 7/28-8/1/75 - Gordon Conference (chaired session on instabilities in layered dichalcogenides)
- 12/3/75 - UC Berkeley (seminar: "Superconducting Power Transmission Lines")
- 2/19/76 - UC Santa Cruz (seminar: "Use of Helium and Applied Superconductivity")
- 2/24/76 - Bell Laboratories (seminar: "The Superconducting Power Transmission Line")
- 3/4/76 - "Distinguished Lecture Series, U New Mexico (Superconductivity: Materials for Energy Use")
- 8/17-20/76 - 1976 Applied Superconductivity Conference (Conference Chairman)

8/14-15/77 International Conference on Physics of Transition Metals,  
Toronto, Canada

11/14/77 General Motors Research Laboratories, Warren, Michigan  
(seminar: "Superconductivity in Transition  
Metals")

J. M. E. Harper

4/7-11/75 IBM, Yorktown Heights, New York

Exxon Research and Engineering Co., Linden, New Jersey

Bell Laboratories, Murray Hill, New Jersey

(seminar: Thermal Properties of Metals with Low  
Temperature Structural Instabilities)

T. J. Jach

IBM, Yorktown Heights, New York

Bell Laboratories, Murray Hill, New Jersey

(seminar: Observation of Nuclear Quadrupole  
Resonance with Superconducting  
Magnetometers)